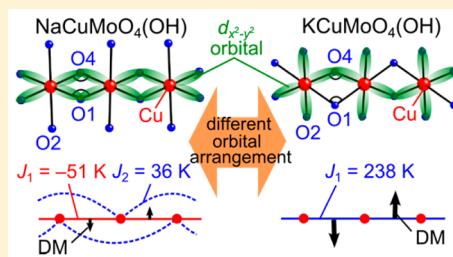


Orbital Arrangements and Magnetic Interactions in the Quasi-One-Dimensional Cuprates $ACuMoO_4(OH)$ ($A = Na, K$)Kazuhiro Nawa,* Takeshi Yajima, Yoshihiko Okamoto,[†] and Zenji Hiroi

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Supporting Information

ABSTRACT: A new spin-1/2 quasi-one-dimensional antiferromagnet $KCuMoO_4(OH)$ is prepared by the hydrothermal method. The crystal structures of $KCuMoO_4(OH)$ and the already-known Na-analogue $NaCuMoO_4(OH)$ are isotypic, comprising chains of Cu^{2+} ions in edge-sharing $CuO_4(OH)_2$ octahedra. Despite the structural similarity, their magnetic properties are quite different because of the different arrangements of $d_{x^2-y^2}$ orbitals carrying spins. For $NaCuMoO_4(OH)$, $d_{x^2-y^2}$ orbitals are linked by superexchange couplings via two bridging oxide ions, which gives a ferromagnetic nearest-neighbor interaction J_1 of -51 K and an antiferromagnetic next-nearest-neighbor interaction J_2 of 36 K in the chain. In contrast, a staggered $d_{x^2-y^2}$ orbital arrangement in $KCuMoO_4(OH)$ results in superexchange couplings via only one bridging oxide ion, which makes J_1 antiferromagnetic as large as 238 K and J_2 negligible. This comparison between the two isotypic compounds demonstrates an important role of orbital arrangements in determining the magnetic properties of cuprates.



INTRODUCTION

Quantum spin systems attract much attention because quantum fluctuations may suppress conventional magnetic order and can induce exotic states such as a spin liquid.^{1,2} The most promising candidates are found in compounds that contain Cu^{2+} ions with spin 1/2. The Cu^{2+} ion has a d^9 electronic configuration with one unpaired electron occupying either the $d_{x^2-y^2}$ or the d_z^2 orbital in the octahedral crystal field. This degeneracy is eventually lifted by lowering symmetry due to the Jahn–Teller effect: when two opposite ligands move away and the other four ligands come closer, the $d_{x^2-y^2}$ orbital is selected to carry the spin, whereas the d_z^2 orbital is selected in the opposite case. Since this Jahn–Teller energy is quite large, either distortion of octahedra always occurs in actual compounds. Since magnetic interactions between neighboring Cu spins are caused by superexchange interactions via overlapping between Cu d orbitals and O ligand p orbitals, the arrangements of the Cu d orbitals and their connections via ligands are important to determine the magnetic properties of cuprates.

Among many cuprates, a rich variety of crystal structures are found in natural minerals and their synthetic analogues. Recently, for example, many copper minerals with Cu^{2+} ions in the kagome geometry have been focused on from the viewpoint of quantum magnetism: Volborthite ($Cu_3V_2O_7(OH)_2 \cdot 2H_2O$),^{3–5} Herbertsmithite ($Zn_{1-x}Cu_{3+x}(OH)_6Cl_2$),⁶ Vesignieite ($BaCu_3V_2O_8(OH)_2$),⁷ and so on. In addition, there are a lot of minerals where Cu^{2+} forms one-dimensional (1D) chains, such as Chalcantite ($CuSO_4 \cdot 5H_2O$),⁸ Linarite ($PbCuSO_4(OH)_2$),⁹ and Natrochalcite ($NaCu_2(SO_4)_2(OH) \cdot H_2O$).¹⁰ Natrochalcite also has many synthetic analogues of $ACu_2(SO_4)_2(OH) \cdot H_2O$ ($A = K, Rb, Ag$)

and $ACu_2(SeO_4)_2(OH) \cdot H_2O$ ($A = Na, K, Rb, Ag, Tl, NH_4$).^{11,12}

In the present study, we focus on two cuprates which crystallize in the Adelite ($CaMgAsO_4(OH)$)–Descloizite ($PbZnVO_4(OH)$) structure type with general formula $ATXO_4(OH)$ ¹³ ($A = Na, Ca, Cd, Pb, \dots$; $T = Mn, Fe, Cu, Zn, \dots$; $X = V, As, Si, Mo, \dots$). This structure contains $TO_4(OH)_2$ octahedra that form chains by edge sharing, and the chains are separated by XO_4 tetrahedra and A atoms. Thus, it is expected that their magnetic properties have 1D character. Although many compounds are known to crystallize in this structure, their magnetic properties have not yet been studied in detail except for $NaCuMoO_4(OH)$.¹⁴ $NaCuMoO_4(OH)$ was first synthesized hydrothermally by Moini et al. in 1986.¹⁵ It crystallizes in an orthorhombic structure with the space group $Pnma$. The Cu^{2+} atom occupies the T site to form a chain made of $CuO_4(OH)_2$ octahedra. Since the $CuO_4(OH)_2$ octahedron is heavily elongated perpendicular to the edge-sharing plaquette of the ligands, the $d_{x^2-y^2}$ orbital lying in the plaquette is selected to carry spin 1/2. Then magnetic interactions between neighboring Cu spins occur by superexchange interactions via two bridging oxide ions. Since the Cu–O–Cu bond angles in this type of chains are approximately 90° , the nearest-neighbor interaction J_1 is expected to be ferromagnetic according to the Goodenough–Kanamori rule. In addition, there is a super-superexchange interaction J_2 through Cu–O–O–Cu paths along the chain, which is antiferromagnetic. In fact, the exchange interactions of $NaCuMoO_4(OH)$ are estimated as $J_1 = -51$ K and $J_2 = 36$ K.¹⁴ Note that the two interactions cause

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frustration in the spin chain, because J_1 favors all spins in the same direction whereas J_2 tends to make every second spin antiparallel. This frustration can lead to a variety of magnetic phases such as spin nematic order, which is theoretically predicted to be a sort of spin liquid with an unusual spin order analogous to the nematic state in liquid crystals^{16–19} but has not yet been observed in actual compounds.²⁰

In this study, we searched for a new compound in this series of compounds and succeeded to synthesize a K analogue, $\text{KCuMoO}_4(\text{OH})$. Its single crystals were grown by the hydrothermal method, and the crystal structure and magnetic properties are investigated. The crystal structures of $\text{KCuMoO}_4(\text{OH})$ and $\text{NaCuMoO}_4(\text{OH})$ are isotypic, but surprisingly, their magnetic properties are totally different. It is shown that the difference comes from a subtle difference in the atomic coordinates, which causes different arrangements of $d_{x^2-y^2}$ orbitals.

EXPERIMENTAL SECTION

Single crystals of $\text{KCuMoO}_4(\text{OH})$ were synthesized by the hydrothermal method. First, 0.4619 g (1.912 mmol) of $\text{Cu}(\text{NO}_3)_2 \cdot 3\text{H}_2\text{O}$ was dissolved in 10 mL of water. Then 9.525 g (40 mmol) of K_2MoO_4 was added, and the solution was stirred until it became homogeneous. The solution was put in a Teflon beaker of 25 mL volume, placed in a stainless steel autoclave, and heated to 433 K and then slowly to 493 K in 240 h. Greenish crystals with a maximum size of $0.6 \times 0.6 \times 0.6 \text{ mm}^3$ were obtained, as shown in Figure 1a. The crystals were

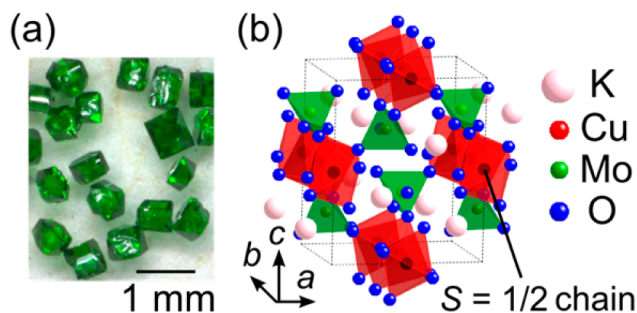


Figure 1. (a) Photograph of single crystals of $\text{KCuMoO}_4(\text{OH})$. (b) Crystal structure of $\text{KCuMoO}_4(\text{OH})$ (hydrogen atoms are omitted). $\text{CuO}_4(\text{OH})_2$ octahedra share their edges to form a chain of $S = 1/2$ along the b axis.

characterized by means of powder X-ray diffraction (XRD) using $\text{Cu K}\alpha$ radiation (Rigaku, RINT-2000), chemical composition analysis using inductively coupled plasma (ICP) spectroscopy (Horiba, JY138 KH), and thermal gravimetry (Bruker AXS TG-DTA2020SAH). A powder XRD pattern from crushed crystals (Figure S1, Supporting Information) shows that all diffraction peaks are indexed based on an orthorhombic structure with the space group $Pnma$, indicating that there is no impurity inclusion. The obtained lattice constants of $a = 7.9901(1) \text{ \AA}$, $b = 6.3201(1) \text{ \AA}$, and $c = 9.5818(1) \text{ \AA}$ are slightly larger than those of the Na compound ($a = 7.726 \text{ \AA}$, $b = 5.968 \text{ \AA}$, and $c = 9.495 \text{ \AA}$ ¹⁵), reflecting the larger ionic radius of the K^+ ion compared with the Na^+ ion. The chemical compositions of K, Mo, and Cu determined by ICP are 13.5(1), 22.0(1), and 34.0(2) wt %, respectively, which are close to the stoichiometric compositions expected for $\text{KCuMoO}_4(\text{OH})$: 14.0, 22.7, and 34.3 wt %. A dehydration reaction was observed upon heating above 623 K with a weight loss of 3.2(1)%, which indicates that 0.5 mol of H_2O per formula unit has been lost as expected from the stoichiometric composition. Therefore, we successfully obtained a new compound $\text{KCuMoO}_4(\text{OH})$, which is obviously related to the Na analogue.

RESULTS AND DISCUSSION

Crystal Structure. To determine the crystal structure, XRD data on a single crystal were collected in a R-Axis RAPID IP diffractometer with a monochromated $\text{Mo K}\alpha$ radiation ($\lambda = 0.71075 \text{ \AA}$) at 296 K. The structure was refined against $|F^2|$ by the direct method using the SHELXL2013 software. A multiscan absorption correction was employed, and the displacement parameters were refined as anisotropic. The H atom could not be located. Structural refinements have converged with a reasonably small R factor of 2.04%; crystal data and atomic parameters are listed in Tables 1 and 2,

Table 1. Crystal Data for $\text{KCuMoO}_4(\text{OH})$

formula	$\text{KCuMoO}_4(\text{OH})$
cryst size	$0.10 \times 0.08 \times 0.05 \text{ mm}^3$
cryst syst	orthorhombic
space group	$Pnma$
a (Å)	7.98385(17)
b (Å)	6.31560(14)
c (Å)	9.5641(4)
V (Å ³)	482.25(3)
Z	4
calcd density (g cm^{-3})	3.851

respectively. Further details of the refinement parameter, selected bond lengths, bond valence sums, and the crystal structure can be obtained from Tables S1, S2, and S3 and the cif file, respectively, which are included in the Supporting Information. Figure 1b shows the crystal structure of the K compound. Basically, it is identical to that of the Na compound, in which $\text{CuO}_4(\text{OH})_2$ octahedra form edge-sharing chains along the b axis, and the chains are separated by the MoO_4 tetrahedra and $\text{K}(\text{Na})$ atoms. However, because of the size difference between K^+ and Na^+ ions, the shapes of the edge-sharing chains are slightly but significantly different between the two compounds, as will be mentioned later.

Magnetic Properties. Despite the structural similarities between the two compounds, their magnetic properties are quite different. Figure 2 shows the temperature dependences of magnetic susceptibility χ measured at a magnetic field of 1 T. The χ of the Na compound increases with decreasing temperature, following a Curie–Weiss (CW) law with a CW temperature of -5.0 K . Then it exhibits a broad peak at 14 K due to an antiferromagnetic short-range order.¹⁴ In sharp contrast, for the K compound, the magnitude of χ is much smaller and the peak temperature lies at a much higher temperature of 150 K. These indicate that there are stronger antiferromagnetic interactions in the K compound. In fact, the CW temperature is estimated to be -205 K for χ in the range of 250–300 K at $H \parallel a$. Note that this CW temperature can be an underestimate because of the limited fitting range which is not high enough compared with the peak temperature. Both the magnitude and the peak temperature of the data in the wide temperature range of 80–300 K are well reproduced by a spin-1/2 1D antiferromagnetic chain model²¹ in which the g factor and the nearest-neighbor interaction J_1 are used as variables together with a temperature-independent contribution χ_0 . The fitting shown by the dashed curve in Figure 2 for $H \parallel a$ is reasonably good and yields $g = 1.92(1)$, $J_1 = 196(1) \text{ K}$, and $\chi_0 = 1.19(5) \times 10^{-4} \text{ cm}^3 \text{ mol}^{-1}$. Thus, a simple spin chain with a large antiferromagnetic interaction may be realized in the K compound, in contrast to the frustrated $J_1 - J_2$ chain with $J_1 =$

Table 2. Atomic Positions and Displacement Parameters for $\text{KCuMoO}_4(\text{OH})$

atom	site	x	y	z	U_{11}	U_{22}	U_{33}	U_{12}	U_{23}	U_{13}
Cu	4b	0.5	0	0	0.0098(3)	0.0072(3)	0.0088(3)	0.00187(14)	−0.00132(13)	−0.00047(15)
K	4c	0.37445(9)	0.25	0.33160(10)	0.0132(4)	0.0189(4)	0.0121(4)	0	0.0019(3)	0
Mo	4c	0.11152(3)	−0.25	0.20063(3)	0.00605(18)	0.00793(18)	0.0083(2)	0	0.00100(9)	0
O(1)	4c	0.2919(3)	−0.25	0.0942(3)	0.0117(12)	0.0176(12)	0.0139(13)	0	0.0029(10)	0
O(2)	8d	0.1039(2)	0.5159(3)	0.30492(19)	0.0139(9)	0.0104(9)	0.0123(10)	−0.0008(6)	0.0028(6)	−0.0001(7)
O(3)	4c	−0.0594(3)	−0.25	0.0869(3)	0.0106(12)	0.0343(16)	0.0173(14)	0	−0.0044(11)	0
O(4)	4c	0.3792(3)	0.25	0.0538(3)	0.0089(11)	0.0095(12)	0.0104(13)	0	−0.0009(8)	0

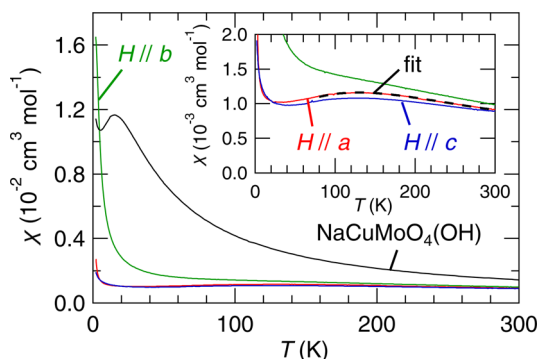


Figure 2. Temperature dependences of magnetic susceptibility measured in a magnetic field of 1 T upon heating after zero-field cooling. Red, green, and blue curves are obtained at $H||a$, b , and c , respectively, for $\text{KCuMoO}_4(\text{OH})$. Black curve represents the χ of $\text{NaCuMoO}_4(\text{OH})$ for comparison;¹⁴ χ of the polycrystalline sample is shown since its anisotropy is small. (Inset) Blowup around the broad maximum. Dashed curve represents a fit to a 1D antiferromagnetic chain model.²¹

−51 K and $J_2 = 36$ K in the Na compound.¹⁴ In addition, χ of the K compound exhibits a large anisotropy below the peak temperature, while the Na compound has little anisotropy in χ ; there is a small anisotropy arising from that of the g factor in the Na compound. χ at $H||b$ is enormously enhanced toward $T = 0$, while those at $H||a$ and $H||c$ decrease, followed by small upturns below ~ 10 K. Note that the χ value at 2 K for $H||b$ becomes 8 times as large as those for $H||a$ and $H||c$. These features are sample independent, suggesting that the origin is not extrinsic from defects or magnetic impurities but intrinsic, which will be discussed later.

Heat capacity measurements also confirm that $\text{KCuMoO}_4(\text{OH})$ is a spin-1/2 1D antiferromagnet. Figure 3 shows a magnetic heat capacity divided by temperature, C_{mag}/T , obtained by subtracting a lattice contribution estimated by the Debye model from the raw data. The C_{mag}/T measured at

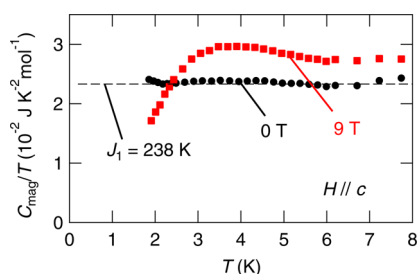


Figure 3. Temperature dependences of magnetic heat capacity divided by temperature in magnetic fields of 0 and 9 T. Dashed line on the zero-field data represents a fit to a 1D antiferromagnetic chain with $J_1 = 238$ K.²¹

zero field does not show any anomaly down to 2 K. Thus, the transition temperature should be lower than 2 K, which is 2 orders of magnitude smaller than the CW temperature, indicating a good one-dimensionality in magnetic interactions. Note that C_{mag}/T becomes almost constant at temperatures below ~ 6 K. Such a T -linear dependence in C_{mag} is not due to electron heat capacity, since this compound is insulating. Alternatively, the T -linear heat capacity is characteristic of a spin-1/2 1D antiferromagnetic chain and is given by $C_{\text{mag}} = 2R/3J_1 \times T$, where R is the gas constant.^{21,22} The observed $C_{\text{mag}}/T = 0.0234(1) \text{ J K}^{-2} \text{ mol}^{-1}$ yields $J_1 = 238(1) \text{ K}$, which is in reasonably good agreement with $J_1 = 196(1) \text{ K}$ from the χ data. We take $J_1 = 238 \text{ K}$ from the heat capacity data to be more reliable, because there are some ambiguities in fitting the χ data. On the other hand, a marked change in C_{mag} is observed when a magnetic field is switched on. Applying 9 T field enhances the C_{mag} above 2.5 K and reduces below 2.5 K, indicating that a significant amount of magnetic entropy has been pushed to higher temperatures. This suggests that an energy gap opens by the magnetic field, as discussed later.

Discussion. It is now clear that the two compounds $\text{NaCuMoO}_4(\text{OH})$ and $\text{KCuMoO}_4(\text{OH})$ are isotypic in the crystal structure but show very different magnetic properties. Here we discuss the magnetic interactions in them and also the unusual magnetic susceptibility of the K compound. As schematically illustrated in Figure 4a, the Na compound has two competing intrachain interactions of $J_1 = -51 \text{ K}$ and $J_2 = 36 \text{ K}$, while the K compound has a much stronger antiferromagnetic interaction of $J_1 = 238 \text{ K}$. This striking difference must be originated from local structures. Let us

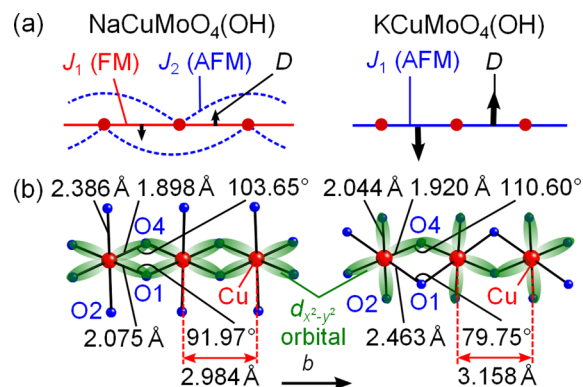


Figure 4. (a) Schematic representations of magnetic interactions present in the chains of $\text{NaCuMoO}_4(\text{OH})$ and $\text{KCuMoO}_4(\text{OH})$. J_1 and J_2 denote nearest-neighbor and next-nearest-neighbor interactions, respectively, and D refers to Dzyaloshinskii–Moriya vector. Abbreviations FM and AFM represent ferromagnetic and antiferromagnetic interactions, respectively. (b) Local structures of the chains made of $\text{CuO}_4(\text{OH})_2$ octahedra and the arrangements of $d_{x^2-y^2}$ orbitals (green lobes).

compare the shapes and linkages of $\text{CuO}_4(\text{OH})_2$ octahedra between the two compounds in Figure 4b; note that all octahedra are crystallographically identical in each compound. There are 3 oxygen sites forming the octahedron: O1 and O4 (probably OH) bridge two neighboring octahedra, and O2 is connected to only one Cu. In the Na compound, the octahedron is elongated toward O2 with a large difference in bond lengths: $d(\text{Cu}-\text{O1}) = 2.075 \text{ \AA}$ and $d(\text{Cu}-\text{O4}) = 1.898 \text{ \AA}$ are much shorter than $d(\text{Cu}-\text{O2}) = 2.386 \text{ \AA}$. Thus, it is no doubt that the $d_{x^2-y^2}$ orbital extending toward O1 and O4 is responsible for $S = 1/2$. On the other hand, in the K compound, the octahedron is elongated toward O1: $d(\text{Cu}-\text{O2}) = 2.044 \text{ \AA}$ and $d(\text{Cu}-\text{O4}) = 1.920 \text{ \AA}$ are much shorter than $d(\text{Cu}-\text{O1}) = 2.463 \text{ \AA}$. This geometry makes a spin-carrying $d_{x^2-y^2}$ orbital extend toward O2 and O4. Therefore, the $\text{CuO}_4(\text{OH})_2$ octahedra are distorted in different ways in the two compounds, resulting in different arrangements of $d_{x^2-y^2}$ orbitals.

It is to be emphasized here that magnetic couplings between Cu spins should critically depend on the arrangements of $d_{x^2-y^2}$ orbitals. J_1 occurs by superexchange interactions via two paths, Cu–O1–Cu and Cu–O4–Cu in the Na compound, but via only one path Cu–O4–Cu in the K compound. These interactions must be sensitive to the bond angles. According to the Goodenough–Kanamori rule, J_1 becomes ferromagnetic when a Cu–O–Cu bond angle is close to 90° and changes into antiferromagnetic when the bond angle exceeds $95-98^\circ$.^{23,24} In the Na compound, the Cu–O1–Cu path with a bond angle of 91.97° must be ferromagnetic, while the Cu–O4–Cu path with a bond angle of 103.65° may be antiferromagnetic: ferromagnetic J_1 may be attained from the dominant contribution of the former path. In contrast, in the K compound, the Cu–O4–Cu path with a large bond angle of 110.60° must make J_1 strongly antiferromagnetic. On the other hand, J_2 is dominated by supersuperexchange interactions via Cu–O1–O4–Cu and Cu–O4–O1–Cu paths, which actually exist in the $d_{x^2-y^2}$ arrangement of the Na compound but not in that of the K compound; little overlap between the $d_{x^2-y^2}$ and the p orbitals of O1 makes J_2 negligible. Therefore, the difference in magnetic interactions in the two compounds is well understood in terms of the local structures.

The different local structures in the two compounds are apparently attributed to the size difference of the Na^+ and K^+ ions. These ions are located between two Cu chains (Figure 1b) and determine some important bond lengths in the chains. In particular, the Cu–Cu distances are significantly different: $d(\text{Cu}-\text{Cu})$ is 2.984 and 3.158 \AA in the Na and K compounds, as shown in Figure 4b. The former is close to those in other compounds having edge-sharing chains where $d_{x^2-y^2}$ orbitals are linked by two oxide atoms as in the Na compound: $d(\text{Cu}-\text{Cu}) = 2.905$ and 2.941 \AA for LiVCuO_4 ²⁵ and CuGeO_3 ,²⁶ respectively. In contrast, the latter long Cu–Cu distance may not be compatible with this type of chains but prefer a different local structure with a large Cu–O–Cu bond angle, as realized in the K compound. Therefore, even in the isotypic structures, the two types of local structures and thus of orbital arrangements are made possible.

Next, we will consider the origin of the unusual behavior in the magnetic susceptibility of $\text{KCuMoO}_4(\text{OH})$: the large enhancement at low temperatures only for $H \parallel b$. Similar behavior has been observed in Cu–benzoate,^{27–29} Yb_4As_3 ,³⁰ and KCuGaF_6 ³¹ and well explained by the presence of staggered Dzyaloshinskii–Moriya (DM) interactions.^{32,33} We

think that a similar mechanism applies to the present compound. The model Hamiltonian considered is given as

$$H = \sum_i \{J_1 \mathbf{S}_i \cdot \mathbf{S}_{i+1} - g\mu_B H S_{iz} + (-1)^i \mathbf{D} \cdot (\mathbf{S}_i \times \mathbf{S}_{i+1})\} \quad (1)$$

where μ_B , H , and D denote the Bohr magneton, magnetic field, and DM vector, respectively. The DM interactions tend to modify a simple antiferromagnetic alignment of spins on a chain; they serve to incline spin \mathbf{S}_{i+1} with respect to spin \mathbf{S}_i . When all DM vectors are identical, spin \mathbf{S}_{i+1} is always inclined in the same direction with respect to spin \mathbf{S}_i so as not to generate uncompensated magnetic moments in an antiferromagnetic chain. However, when DM vectors are staggered in orientation as depicted in Figure 4a, the direction of this inclination alternates so as to generate net moments. In other words, staggered DM interactions in an antiferromagnetic chain classically stabilize two-sublattice-canted antiferromagnetism. This leads to a large increase in χ at low temperatures below a broad maximum, where an antiferromagnetic short-range order develops, even in a paramagnetic state as observed for the K compound. Moreover, this enhancement should occur effectively when H is perpendicular to D and be absent for $H \parallel D$, because spins are inclined to directions perpendicular to magnetic field for $H \parallel D$. The observed anisotropic behavior for the K compound, a large enhancement in χ only for $H \parallel b$, is understood if the D is perpendicular to the b axis.

Generally speaking, a possible direction of D is restricted by crystal symmetry. In the Na and K compounds, there is a mirror plane that bisects the Cu–Cu bond and a 2-fold screw axis that passes through the Cu sites. The former requires D to align perpendicular to the chain ($D_b = 0$), and the latter requires D to alternate along the chain direction ($D_{i,a} = -D_{i+1,a}$, $D_{i,c} = -D_{i+1,c}$). Thus, the DM interactions are completely staggered with the DM vectors always perpendicular to the b axis in both compounds, as depicted in Figure 4a. Concerning the magnitude of D , one expects that it is larger in the K compound because the alternation of the $d_{x^2-y^2}$ orbital along the chain is absent in the Na compound but present in the K compound. This is the reason why an enhancement of χ is prominent in the K compound.

It is also characteristic for a spin chain with staggered DM interactions that an energy gap is induced by an applied magnetic field. In fact, such a gap opening has been extensively studied in Cu–benzoate through heat capacity,²⁹ inelastic neutron scattering^{28,29} and ESR experiments.³⁴ Thus, the field-induced energy gap observed in the heat capacity data of the K compound also supports the presence of staggered DM interactions. More detailed analyses on the magnetic properties will be given elsewhere.

The dramatic interplay between crystal structures and magnetic interactions observed in $\text{NaCuMoO}_4(\text{OH})$ and $\text{KCuMoO}_4(\text{OH})$ is interesting and unique. A similar observation is found in related compounds such as CuCl_2 ,^{35–38} $\text{CuCl}_2 \cdot 2\text{H}_2\text{O}$,^{37,39,40} and $\text{CuCl}_2 \cdot 2\text{NC}_5\text{H}_5$,^{41,42} which contains spin chains of edge-sharing CuCl_4X_2 ($\text{X} = \text{Cl}$, O , and N , respectively) octahedra. In these compounds, the different ligands may induce changes in the orbital arrangement. On one hand, “orbital switching” from the d_z^2 to the $d_{x^2-y^2}$ orbital observed in Volborthite, which is probably caused by a change in the hydrogen bonding of water molecules,^{4,5} gives rise to a substantial change in magnetic interactions. However, these phenomena are essentially related to the changes of crystal symmetry. In sharp contrast, in the present

compounds, only the small differences of atomic coordinates under the same crystal symmetry lead to completely different d orbital arrangements and thus cause the large changes in the sign and magnitude of magnetic interactions. In future work, we would like to synthesize solid solutions between the two compounds to control the magnetic interactions systematically.

CONCLUSIONS

We prepared single crystals of a new quasi-1D cuprate $\text{KCuMoO}_4(\text{OH})$ and investigated its crystal structure and magnetic properties in comparison with those of the Na analogue $\text{NaCuMoO}_4(\text{OH})$. The two compounds are structurally isotypic but have different arrangements of $d_{x^2-y^2}$ orbitals. As the result, the magnetic properties become critically different: $\text{NaCuMoO}_4(\text{OH})$ provides a frustrated $J_1 - J_2$ chain system, while $\text{KCuMoO}_4(\text{OH})$ is a 1D antiferromagnet with $J_1 = 238$ K with staggered DM interactions. An interesting relationship between crystal structures and magnetic interactions is demonstrated by the comparison of the two compounds.

ASSOCIATED CONTENT

Supporting Information

Powder XRD diffraction pattern, refinement parameters, selected bond lengths, bond valence sums, and .cif file representing the refined structure of $\text{KCuMoO}_4(\text{OH})$. The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.inorgchem.5b00686.

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Notes

The authors declare no competing financial interest.

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REFERENCES

- (1) Mikeska, H.-J.; Kolezhuk, A. K. One-dimensional magnetism. In *Quantum Magnetism*; Schollwöck, U., Richter, J., Farnell, D. J. J., Bishop, R. F., Eds.; Lecture Notes in Physics; Springer: Berlin, 2004; Vol. 645, pp 1–83.
- (2) Balents, L. *Nature* **2010**, 464, 199–208.
- (3) Hiroi, Z.; Hanawa, M.; Kobayashi, N.; Nohara, M.; Takagi, H.; Kato, Y.; Takigawa, M. *J. Phys. Soc. Jpn.* **2001**, 70, 3377–3384.
- (4) Yoshida, H.; Yamaura, J.; Isobe, M.; Okamoto, Y.; Nilsen, G. J.; Hiroi, Z. *Nat. Commun.* **2012**, 3, 860.
- (5) Ishikawa, H.; Yamaura, J.; Okamoto, Y.; Yoshida, H.; Nilsen, G. J.; Hiroi, Z. *Acta Crystallogr., Sect. C* **2012**, C68, i41–i44.
- (6) Shores, M. P.; Nytko, E. A.; Bartlett, B. M.; Nocera, D. G. *J. Am. Chem. Soc.* **2005**, 127, 13462–13463.
- (7) Okamoto, Y.; Yoshida, H.; Hiroi, Z. *J. Phys. Soc. Jpn.* **2009**, 78, 033701.
- (8) Geballe, T. H.; Giaque, W. F. *J. Am. Chem. Soc.* **1952**, 74, 3513–3519.
- (9) Baran, M.; Jedrzejczak, A.; Szymczak, H.; Maltsev, V.; Kamieniarz, G.; Szukowski, G.; Loison, C.; Ormeci, A.; Drechsler, S.-L.; Rosner, H. *Phys. Status Solidi* **2006**, 1, 220–224.

- (10) Takeda, S.; Watanabe, A.; Maruta, G.; Matsuo, T. *Mol. Cryst. Liq. Cryst.* **2002**, 376, 443–448.
- (11) Giester, G.; Zemmann, J. *Z. Kristallogr.* **1987**, 179, 431–442.
- (12) Giester, G. *Z. Kristallogr.* **1989**, 187, 239–247.
- (13) Palache, C.; Berman, H.; Frondel, C. *The System of Mineralogy of J. D. Dana and E. S. Dana, Yale University 1837–1890*; John Wiley and Sons, Inc.: New York and London, 1951; Vol. 2, p 804.
- (14) Nawa, K.; Okamoto, Y.; Matsuo, A.; Kindo, K.; Kitahara, Y.; Yoshida, S.; Ikeda, Y.; Hara, S.; Sakurai, T.; Okubo, S.; Ohta, H.; Hiroi, Z. *J. Phys. Soc. Jpn.* **2014**, 83, 103702.
- (15) Moini, A.; Peascoe, R.; Rudolf, P. R.; Clearfield, A. *Inorg. Chem.* **1986**, 25, 3782–3785.
- (16) Kecke, L.; Momoi, T.; Furusaki, A. *Phys. Rev. B* **2007**, 76, 060407.
- (17) Vekua, T.; Honecker, A.; Mikeska, H.-J.; Heidrich-Meisner, F. *Phys. Rev. B* **2007**, 76, 174420.
- (18) Hikihara, T.; Kecke, L.; Momoi, T.; Furusaki, A. *Phys. Rev. B* **2008**, 78, 144404.
- (19) Sudan, J.; Lüscher, A.; Läuchli, A. M. *Phys. Rev. B* **2009**, 80, 140402.
- (20) Büttgen, N.; Nawa, K.; Fujita, T.; Hagiwara, M.; Kuhns, P.; Prokofiev, A.; Reyes, A. P.; Svistov, L. E.; Yoshimura, K.; Takigawa, M. *Phys. Rev. B* **2014**, 90, 134401.
- (21) Johnston, D. C.; Kremer, R. K.; Troyer, M.; Wang, X.; Klümper, A.; Bud'ko, S. L.; Panchula, A. F.; Canfield, P. C. *Phys. Rev. B* **2000**, 61, 9558–9606.
- (22) Bonner, J. C.; Fisher, M. E. *Phys. Rev.* **1964**, 135, A640–A658.
- (23) Crawford, V. H.; Richardson, H. W.; Wasson, J. R.; Hodgson, D. J.; Hatfield, W. E. *Inorg. Chem.* **1976**, 15, 2107–2110.
- (24) Mizuno, Y.; Tohyama, T.; Maekawa, S.; Osafune, T.; Motoyama, N.; Eisaki, H.; Uchida, S. *Phys. Rev. B* **1998**, 57, 5326–5335.
- (25) Lafontaine, M. A.; Leblanc, M.; Ferey, G. *Acta Crystallogr., Sect. C* **1989**, C45, 1205–1206.
- (26) Völlenkle, H.; Wittmann, A.; Nowotny, H. *Montash. Chem.* **1967**, 98, 1352–1357.
- (27) Koizumi, H.; Osaki, K.; Watanabe, T. *J. Phys. Soc. Jpn.* **1963**, 18, 117–124.
- (28) Dender, D. C.; Davidović, D.; Reich, D. H.; Broholm, C.; Lefmann, K.; Aeppli, G. *Phys. Rev. B* **1996**, 53, 2583–2589.
- (29) Dender, D. C.; Hammar, P. R.; Reich, D. H.; Broholm, C.; Aeppli, G. *Phys. Rev. Lett.* **1997**, 79, 1750–1753.
- (30) Kohgi, M.; Iwasa, K.; Mignot, J.-M.; Fåk, B.; Gegenwart, P.; Lang, M.; Ochiai, A.; Aoki, H.; Suzuki, T. *Phys. Rev. Lett.* **2001**, 86, 2439–2442.
- (31) Morisaki, R.; Ono, T.; Tanaka, H.; Nojiri, H. *J. Phys. Soc. Jpn.* **2007**, 76, 063706.
- (32) Oshikawa, M.; Affleck, I. *Phys. Rev. Lett.* **1997**, 79, 2883–2886.
- (33) Affleck, I.; Oshikawa, M. *Phys. Rev. B* **1999**, 60, 1038–1056.
- (34) Nojiri, H.; Ajiro, Y.; Asano, T.; Boucher, J.-P. *New J. Phys.* **2006**, 8, 218.
- (35) de Haas, W. J.; Gorter, C. J. *CComm. Kamerlingh Onnes Lab. Univ. Leiden Suppl.* **1931**, 215a.
- (36) Starr, C.; Bitter, F.; Kaufmann, A. R. *Phys. Rev.* **1940**, 58, 977–983.
- (37) Schmitt, M.; Janson, O.; Schmidt, M.; Hoffmann, S.; Schnelle, W.; Drechsler, S.-L.; Rosner, H. *Phys. Rev. B* **2009**, 79, 245119.
- (38) Banks, M. G.; Kremer, R. K.; Hoch, C.; Simon, A.; Ouladdiaf, B.; Broto, J.-M.; Rakoto, H.; Lee, C.; Whangbo, M.-H. *Phys. Rev. B* **2009**, 80, 024404.
- (39) Handel, J. v. d.; Gijsman, H. M.; Poulsen, N. J. *Physica* **1952**, 18, 862–876.
- (40) Marshall, W. J. *Phys. Chem. Solids* **1958**, 7, 159–164.
- (41) Dunitz, J. D. *Acta Crystallogr.* **1957**, 10, 307–313.
- (42) Takeda, K.; Matsukawa, S.; Haseda, T. *J. Phys. Soc. Jpn.* **1971**, 30, 1330–1336.